

- GLUSKER, J., PATTERSON, B. & ROSSI, M. (1987). *Patterson and Pattersons. Fifty Years of the Patterson Function*. IUCr/Oxford Univ. Press.
- HARKER, D. & KASPER, J. S. (1948). *Acta Cryst.* **1**, 70–75.
- HAUPTMAN, H. (1991). *Rep. Prog. Phys.* **54**, 1427–1454.
- IGLESIAS, J. (1981). *Z. Kristallogr.* **156**, 187–196.
- IRELAND, K. & ROSEN, M. (1982). *A Classical Introduction to Modern Number Theory*. New York/Heidelberg/Berlin: Springer-Verlag.
- KARLE, J. & HAUPTMAN, H. (1950). *Acta Cryst.* **3**, 181–187.
- KLUG, A. (1958). *Acta Cryst.* **11**, 515–543.
- LENSTRA, H. W. (1993). *Acta Arith.* **64**, 383–388.
- LEVEQUE, W. J. (1956). *Topics in Number Theory*. Reading, MA: Addison-Wesley.
- OXTOBY, J. (1987). In *Patterson and Pattersons. Fifty Years of the Patterson Function*, edited by J. GLUSKER, B. PATTERSON & M. ROSSI. IUCr/Oxford Univ. Press.
- PATTERSON, L. (1934). *Phys. Rev.* **46**, 372–376.
- PATTERSON, L. (1935). *Z. Kristallogr.* **90**, 517–542.
- PATTERSON, L. (1939). *Phys. Rev.* **15**, 682.
- PATTERSON, L. (1944). *Phys. Rev.* **65**, 195–201.
- PAULING, L. & SHAPPEL, M. (1930). *Z. Kristallogr.* **75**, 128–142.
- PICCARD, S. (1939). *Mem. Univ. Neuchâtel*, No. 13. Paris: Librairie Gauthier-Villars.
- ROSENBLATT, J. (1984). *Commun. Math. Phys.* **95**, 317–343.
- SAYRE, D. (1952). *Acta Cryst.* **5**, 60–65.
- WILSON, A. (1949). *Acta Cryst.* **2**, 318–321.
- ZACHARIASEN, W. (1928). *Z. Kristallogr.* **67**, 455–464.

Acta Cryst. (1995). **A51**, 323–328

Theoretical *Ab Initio* Calculations of the Structure Factors of Fluorite (CaF₂)

BY ALBERT LICHANOT AND MICHEL RÉRAT

Laboratoire de Chimie Structurale URA 474, Université de Pau, IFR, rue J. Ferry, 64000 Pau, France

AND MICHELE CATTI

Dipartimento di Chimica Fisica ed Elettrochimica, Università di Milano, via Golgi 19, I-20133 Milano, Italy

(Received 25 July 1994; accepted 4 November 1994)

Abstract

Ab initio calculations of static structure factors of fluorite (CaF₂) are performed by a linear combination of atomic orbitals Hartree–Fock method as implemented in the *CRYSTAL* program. The effect of thermal motion is then introduced by taking into account the atomic mean square displacements given in the literature at different temperatures and leads to dynamic structure factors. Finally, a very slight displacement of fluorine ions with respect to their ideal position is considered, to simulate an anharmonic vibration or disordered structure so as to improve the agreement with experimental data.

Introduction

In a recent theoretical study of fluorite (CaF₂), Catti, Dovesi, Pavese & Saunders (1991) perfected and used an atomic orbitals (AO) basis set for calcium and fluorine, which allowed them to calculate and compare successfully with experiment some ground-state properties: lattice parameter, binding energy, electronic and band structures, and elastic constants. The fully ionic nature of fluorite is clearly shown.

In the present work, we have calculated the static structure factors of fluorite from the wave functions and density matrix obtained by *CRYSTAL*, an *ab initio* periodic Hartree–Fock program (Dovesi, Pisani, Roetti,

Casà & Saunders, 1989; Dovesi, Roetti & Saunders, 1992) by using the same all-electron basis set and computational parameters as those previously defined by Catti *et al.* (1991).

A perturbation of the Fock matrix by thermal motion, assuming that atomic displacements are independent and that the atomic orbitals follow nuclear movements (Azavant, Lichanot, Rérat & Chaillet, 1994), allows us to calculate dynamic structure factors at any temperature by introducing the thermal mean square amplitudes of the atoms given in the literature. Thus, it becomes possible to compare theoretical and experimental values, provided that the latter have been corrected for secondary factors (scale, Lorentz, polarization, absorption, extinction, anomalous scattering, thermal diffuse scattering). For comparison with our calculations, we have considered all the experimental data sets obtained for fluorite in the last 30 years since Togawa's (1964) work. They include: (i) the measurements obtained on the same crystal (American Crystallographic Association single-crystal intensity project) by Abrahams *et al.* (1967) and their assessment and analysis done respectively by Mackenzie & Maslen (1968) and by Cooper (1970); (ii) the data obtained by Zachariasen (1968) with a small spherical crystal; (iii) the data set collected from a crystal of 90 µm by Bachmann, Kohler, Schulz & Weber (1985). These authors have also obtained intensities with synchrotron radiation for the same crystal and for a

smaller one of 6 μm . The study of microcrystals ($2\ \mu\text{m}^3$) performed by Rieck, Euler & Schulz (1988) with synchrotron X-ray diffraction allowed the collection of extinction-free data at short wavelengths for use in high-accuracy crystal structure analysis, as shown by Rieck, Schulz & Siedel (1991) and Cheetham & Wilkinson (1992).

As we see, the data are numerous but not always very accurate. Those collected with a conventional X-ray tube by Abrahams *et al.* (1967) are considered to be outside the range of small $\sin \theta/\lambda$ values indicated by Mackenzie & Maslen (1968) and analysed by Cooper (1970) to obtain very valid Debye–Waller factors. Those collected with synchrotron radiation and microcrystals concern on the contrary the small $\sin \theta/\lambda$ values and lead to abnormally high Debye–Waller factors (Rieck *et al.*, 1988). It is therefore useful and interesting to present a theoretical study of the structure factors of fluorite over a large range of $\sin \theta/\lambda$ values to $1.5\ \text{\AA}^{-1}$ with the same precision for the small and high angles.

The structure of this paper is as follows: in §I, the static structure factors are calculated and analysed *versus* the deformation of the ions. §II corresponds to the comparison between the room-temperature experimental data and our dynamic structure factors calculated with the same atomic mean square amplitudes. The variations of the structure factors with temperature and position of fluorine in the cell are also studied.

I. Static structure factors

The method of calculation and the description of the atomic orbitals basis set are already published by Catti *et al.* (1991). It is recalled that we only use the all-electron *ab initio* Hartree–Fock computational scheme described by Pisani, Dovesi & Roetti (1988) and implemented in the *CRYSTAL* program (Dovesi *et al.*, 1989) and that the two basis sets are 8-6-511G* for calcium and 7-311G for fluorine, respectively: the star indicates the presence of *d* polarization functions in the calcium basis set represented by a three-Gaussian-functions contraction.

CaF_2 is cubic *Fm3m* with four formula units in the cell. The lattice parameter adopted in these calculations is the experimental one extrapolated to $T = 0\ \text{K}$ ($5.445\ \text{\AA}$), as deduced from the thermal expansion given by White (1980) and from the unit-cell edge at 300 K ($a = 5.463\ \text{\AA}$; Wyckoff, 1968). With these data, the wave function and electron density have been calculated.

To illustrate these calculations, we have chosen to present shell Mulliken populations and to compare them with those of the isolated ions described with the same basis sets. As Table 1 shows, the classical situation of a fully ionic compound is found with a net charge of $0.934e^-$ on fluorine. A significant contraction of the anion induced by the crystalline field of the neighbouring cations is also observed: the Mulliken population of the most diffuse function of fluorine in CaF_2 is smaller by

Table 1. Atomic Mulliken populations per valence-electron shell in CaF_2 and in the isolated ions (*f*) described by the same basis sets

	2sp	3sp	4sp	5sp	3d	Total
Ca in CaF_2	8.078	4.562	2.519	0.870	0.103	16.132
F in CaF_2	3.933	2.898	1.103			7.934
Ca^{2+} (<i>f</i>)	8.081	4.560	2.533	0.826		16.000
F^- (<i>f</i>)	3.919	2.702	1.380			8.000

$0.28e^-$ than that of free F^- . This result is also shown by Catti *et al.* (1991) from total and differential electron charge density maps.

From the electron density $\rho(\mathbf{r})$, the scattering factor is deduced according to

$$f(\mathbf{s}) = \int \rho(\mathbf{r}) \exp(-i\mathbf{s} \cdot \mathbf{r}) d\mathbf{r}, \quad (1)$$

where \mathbf{r} and \mathbf{s} are the electron position and the scattering vectors, respectively, with

$$\rho(\mathbf{r}) = \sum_{\mu, \nu, \mathbf{g}} P_{\mu\nu}^{\mathbf{g}} \chi_{\mu}^0(A, \mathbf{r}) \chi_{\nu}^{\mathbf{g}*}(B, \mathbf{r}); \quad (2)$$

$\chi_{\mu}^0(A, \mathbf{r})$ is the μ th AO on atom *A* in the reference zero cell, $\chi_{\nu}^{\mathbf{g}}(B, \mathbf{r})$ is the ν th AO on atom *B* in the crystal cell

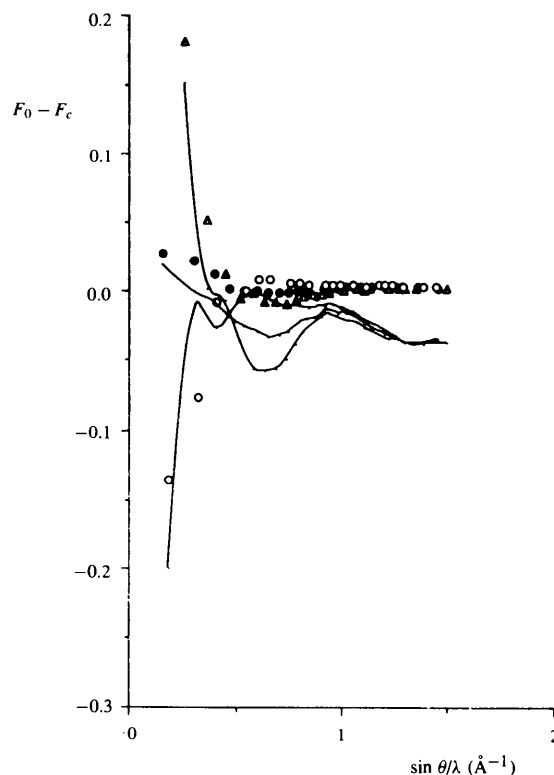


Fig. 1. Differences $F_0 - F_c$ versus $\sin \theta/\lambda$. F_0 values are given in Table 2. Symbols \bullet for odd h, k, l , Δ for $h+k+l = 4n$ and \circ for $h+k+l = 4n+2$ correspond to the differences where F_c values are calculated from our AO basis set. The full lines correspond to the differences where F_c values are calculated from *International Tables for Crystallography* (1992).

the two types of basis sets is observed. As this difference is approximately the same for all $\sin \theta/\lambda$ values, and the f_F^- calculated with our basis set are practically equal to those reported in *International Tables for Crystallography*, it is clear that this deviation is only attributable to a richer description of the core electrons of the calcium in our basis set than in the one used in *International Tables for Crystallography*. To confirm this result in part, calculations of the scattering factor of the isolated calcium ion have been done with our atomic orbital basis set without the $3d$ polarization function. Differences of 0.035 and of -0.02 are observed for the first two reflections and for the three following ones, respectively: these are the only significant differences between the scattering factor $f_{Ca^{2+}}$ values calculated with and without polarization functions.

II. Dynamic structure factors

To calculate structure factors at any temperature, the atomic thermal motion described by mean square displacements $\langle u^2 \rangle$ is introduced by affecting the couples of atomic orbitals (μ, ν) with which the elements of the density matrix are associated. This is done in the context of the Debye hypothesis where atoms are in mutually independent vibration. The distribution probability law $p(\mathbf{u})$ of finding an atom with a displacement \mathbf{u} given by Willis & Pryor (1975) is adopted, and the AOs are assumed to follow the movements of the nuclei. This method leads to an expression of the dynamic structure factor F_T formally identical to the static factor F_0 , with new exponents α'_i of the Gaussian functions in the AOs defined with respect to the static ones (α) by $\alpha'_i = \alpha/(1 + 2\alpha B_{ii})$; B is the tensor of mean square displacement $\langle u^2 \rangle$ and $i (= 1, 2, 3)$ denotes each of the three Cartesian directions (Azavant *et al.*, 1994).

Comparison with the experiment

The structure factors are calculated at room temperature by using the atomic mean square displacements given in the experimental papers and are compared with the corresponding measured values corrected for the anomalous dispersion (*International Tables for Crystallography*, 1992). The agreement factor R is defined as

$$R = \frac{\sum_{h,k,l} |F_T - F_e|}{\sum_{h,k,l} F_e},$$

where F_e represents the experimental value of the structure factor for the hkl reflection and F_T is our calculated value. Table 3 reports the R values obtained for eight data sets denoted ACA, Z_{Mo} , Z_{Cu} , B_I , B_{II} , B_{III} , B_{IV} and R_i . In the American Crystallographic Association (ACA) data (Abrahams *et al.*, 1967), we have only considered those of experiment 2 for which the extinction factor is precisely given by Cooper (1970); Z_{Mo} and Z_{Cu} are symbols associated with the results

Table 3. Agreement factor R between our theoretical structure factors and experimental values obtained at room temperature for n independent reflections

The atomic mean square displacements B (in \AA^2) used in our calculations are those optimized by the authors. Symbols are explained in the text.

	ACA	Z_{Mo}	Z_{Cu}	B_I	B_{II}	B_{III}	B_{IV}	R_i
B_{Ca}	0.53	0.60	0.60	0.38	0.38	0.64	0.61	1.50
B_F	0.78	0.80	0.80	0.83	0.83	0.92	0.81	2.60
R (%)	18	2.5	3	2.7	6.4	8.9	4.6	6.7
	(16)			(1.8)	(2.8)	(2.9)	(2)	(3.6)
n	36	34	11	16	19	41	46	9
	(28)			(15)	(16)	(37)	(42)	(7)

obtained by Zachariassen (1968) for $MoK\alpha$ and $CuK\alpha$ radiations, respectively; B_I , B_{II} , B_{III} and B_{IV} are symbols corresponding to the data of Bachmann *et al.* (1985) obtained with a $6\ \mu\text{m}$ crystal and 90% polarization, a $6\ \mu\text{m}$ crystal and 94% polarization, a $90\ \mu\text{m}$ crystal and 50% polarization and a $90\ \mu\text{m}$ crystal and 0% polarization (conventional X-ray tube); R_i is the symbol associated with the data of Rieck *et al.* (1988). As an example, the experimental values B_{IV} of Bachman *et al.* (1985) corrected with an extinction factor and for anomalous dispersion (*International Tables for Crystallography* 1992) are given in column 7 of Table 2 for comparison with our results (column 5 or 6 of the same table).

Generally speaking, the comparison between our theoretical and the measured structure factors sometimes shows (just as that of the different experimental data sets) very large differences in the group of the eight lowest-angle reflections which contains the five strongest and also the two smallest F values. If a few (from one in B_I to four in B_{III} or B_{IV}) characterized by these large differences are omitted, the number of independent data given in parentheses in Table 3 is obtained, and the corresponding R value is reported in parentheses. Examination of the R values shows that the agreement between our theoretical structure factors and the experimental ones is satisfactory if we exclude the ACA data for which several reasons have been already given by Mackenzie & Maslen (1968) and Cooper (1970) to explain the disagreement. It is convenient to notice that the R values corresponding to the Z_{Cu} , B_I , B_{II} and R_i experiments actually underestimate the agreement, since the data are not very numerous and occur in the range of the smaller $\sin \theta/\lambda$ values, which is affected by the largest errors.

Thermal variations of the structure factors: comparison with the Debye-Waller model

The previous comparison between our theoretical data and the experimental ones at room temperature allows us to calculate accurate and homogeneous values of the structure factors over a large range of $\sin \theta/\lambda$ corresponding to 69 investigated reflections.

Column 5 of Table 2 reports our results at 300 K (F_{300}) while, in column 4, the values of the structure factors F_{300}^{DW} obtained within the Debye–Waller model are given. For these calculations, we have adopted the experimental lattice parameter at room temperature $a_{300} = 5.463 \text{ \AA}$ (Wyckoff, 1968) and the more recent X-ray values of $B_{Ca} = 0.61$ and $B_F = 0.81 \text{ \AA}^2$ (Bachman *et al.*, 1985), which are accurate and in good agreement with other X-ray values (see Table 3) in the range of the small values of $\sin \theta/\lambda$. Comparison between the values of these two columns allows us to show the effect of electron density deformation, which is only taken into account in the F_{300} values. Examination of Table 2 shows differences between F_{300} and F_{300}^{DW} values smaller than 0.5%, proving that fluorite is a nearly fully ionic compound. However, for the reflections 200 and 222 belonging to the even-index group $h + k + l = 4n + 2$ for which small structure-factor values result from the competition between calcium and fluorine components, much larger differences are found, 55 and 4%, respectively. Although these values are the most sensitive to the precision of the calculations, they show the overlap and/or contraction effect of the valence-electron clouds of fluorine and calcium: this result confirms the shape of the static corresponding curve ($h + k + l = 4n + 2$) of Fig. 1 in the range of the smallest $\sin \theta/\lambda$ values. Other compounds with face-centred-cubic structure and a large electronic overlap between neighbouring atoms, such as silicon, present the same behaviour for the 222 reflection (Azavant *et al.*, 1994; Spackman, 1986; Pisani, Dovesi & Orlando, 1992). For a fully ionic compound such as magnesium oxide, the values of F_{300} and F_{300}^{DW} are the same for this reflection (Azavant *et al.*, 1994), indicating no contraction of the electron density.

Structure factors have also been calculated at three temperatures, 50, 300 and 400 K, for which the experimental lattice parameters $a_{50} = 5.446$, $a_{300} = 5.463$ and $a_{400} = 5.475 \text{ \AA}$ are deduced from a_{300} (Wyckoff, 1968) with the help of the data of White (1980) and of Schumann & Neumann (1984). In these calculations, we have adopted the atomic mean square displacements given by Elcombe & Pryor (1970): these latter values are not X-ray results but come from model calculations based indirectly on phonon dispersion measurements. From the obtained values, not given in this paper, the structure factor is a decreasing function *versus* temperature for all reflections except 200 and 222. For the reflections corresponding to odd h , k , l and $h + k + l = 4n$, which have large and more accurate structure factors, the curves $\ln(F_T/F_0) = f(\sin \theta/\lambda)^2$ at a given temperature are linear and the slopes allow us to calculate a Debye–Waller factor B . For odd h , k , l , the values obtained at 50, 300 and 400 K are $B_{50} = 0.186$, $B_{300} = 0.516$ and $B_{400} = 0.673 \text{ \AA}^2$, respectively, and represent the B displacements associated with calcium. These values are, of course, equal to those of Elcombe &

Pryor (1970) used to calculate F_T . For the reflections corresponding to $h + k + l = 4n$, the slopes lead to $B_{50} = 0.233$, $B_{300} = 0.589$ and $B_{400} = 0.763 \text{ \AA}^2$, values that represent rather well a mean Debye–Waller factor B defined by $B = (m_F B_F + m_{Ca} B_{Ca}) / (m_F + m_{Ca})$, where m_F and m_{Ca} are the fluorine and calcium atomic weights, respectively. These results show that at the three studied temperatures the F_T values are similar to those corresponding to the Debye–Waller model and thus confirms the fully ionic character of fluorite and therefore the absence of appreciable overlap between electron clouds.

Anharmonic vibration or disordered structure of fluorine

All the calculations presented so far correspond to the ideal structure where the Ca, F and F' sites are (0,0,0), $(\frac{1}{4}, \frac{1}{4}, \frac{1}{4})$, $(\frac{3}{4}, \frac{3}{4}, \frac{3}{4})$, respectively. In the case of fluorite structure, however, it is known that the Bragg intensities measured by neutron (Willis, 1965) or X-ray (Abrahams *et al.*, 1967) diffraction for reflections occurring at the same value of $\sin \theta/\lambda$ are significantly different (*e.g.* reflections 933, 771, 755). Such a result can be explained by an anharmonic vibration or a disordered structure of fluorine (Willis, 1965).

In this structure, the symmetry of the fluorine site is indeed $43m$ and the absence of a centre of symmetry makes anharmonic vibrations possible. Theoretically, the addition of a cubic term in the development of the crystal potential for the anion and the substitution of the new potential expression into the anion temperature factor allow the modification of the harmonic term of this factor and this result to be accounted for (Willis & Pryor, 1975). Within the group of odd-index reflections sharing the same value of $\sin \theta/\lambda$, the stronger reflection (755) has Miller indices of the type $h + k + l = 4n + 1$ while the weaker reflections (933, 771) correspond to the $h + k + l = 4n - 1$ group.

In the CaF_2 case at room temperature, the neutron intensities of the Bragg reflections lead to $F_{755}/F_{933} = 1.12$ (Willis, 1965) while this ratio is equal to 1.013(16) when calculated from X-ray data (Abrahams *et al.*, 1967; Zachariasen, 1968; Bachman *et al.*, 1985). To explain these results and the absence of detectable splitting of the structure factors in even-index groups with the same $\sin \theta/\lambda$, an anharmonic potential parameter β_F is introduced in the fluorine temperature factor. Its value would be very small and negative ($\beta_F \simeq -4 \times 10^9 \text{ J}^{-3}$) according to Cooper (1970) and does not contribute to modify the B_{Ca} and B_F values.

To simulate the anharmonic vibration of fluorine with the CRYSTAL program, it is possible to move this atom along the [111] direction by a shift δ_F with respect to its ideal position. The modified arrangement of F and F' sites $(\frac{1}{4} + \delta)$, \dots , still related by a centre of inversion, is optimized in order to obtain $F_{II}/F_I = 1.013$ with $B_{Ca} = 0.61$ and $B_F = 0.81 \text{ \AA}^2$ (F_I and F_{II} are the

Table 4. Structure factors F calculated at 300 K with $\delta = -0.0008$ for odd h, k, l reflections having the same value of $\sin \theta/\lambda$.

F_I and F_{II} correspond to $h+k+l = 4n-1$ and $4n+1$, respectively. The values in parentheses correspond to the F_{II}/F_I ratio of the B_{IV} experiment.

$\sin \theta/\lambda$ (\AA^{-1})	0.4756	0.6536	0.7030	0.7926	0.9107	0.9467	1.1395
$h+k+l = 4n-1$					933		
hkl	511	551	731	555	771	951	11,53
F_I	29.564	21.488	19.784	16.916	13.520	12.556	8.072
$h+k+l = 4n+1$							
hkl	333	711	553	751	755	773	975
F_{II}	29.848	21.700	20.012	17.128	13.708	12.732	8.212
F_{II}/F_I	1.010 (0.994)	1.010 (1.006)	1.011 (1.002)	1.012 (1.020)	1.014 (1.001)	1.014 (1.000)	1.017 (-)

structure factors associated with odd-index groups with $h+k+l = 4n-1$ and $4n+1$, respectively). The number of symmetry operators is reduced from 48 to 12 (space group $R\bar{3}m$) and the δ_F value thus obtained is $-8 \times 10^{-4} \text{\AA}$. This value is very small and negative, meaning a displacement of the fluorine towards the neighbouring calcium in the [111] direction. With this value, all the structure factors of the odd-index groups with the same $\sin \theta/\lambda$ value corresponding to $h+k+l = 4n+1$ are stronger than those corresponding to $h+k+l = 4n-1$ and the structure factors of the even-index groups with the same $\sin \theta/\lambda$ value are not split, confirming all the experimental data. The structure factors $F_{300}(\delta)$ calculated with $\delta_F = -8 \times 10^{-4} \text{\AA}$ and the ratios F_{II}/F_I calculated for a few odd-index groups are given in column 6 of Table 2 and in Table 4, respectively. The experimental F_{II}/F_I values of Bachman *et al.* (1985) are given in parentheses in this table for comparison. Examination of Tables 2 and 4 shows that lowering the site symmetry of fluorine by a small shift towards Ca along the [111] direction improves the agreement with X-ray data. This would imply, however, a positive rather than negative sign of β_F according to Willis & Pryor (1975).

Concluding remarks

The *ab initio* LCAO Hartree-Fock approach allowed us to calculate accurate structure factors of fluorite over a large range of $\sin \theta/\lambda$ values. Perturbation of the static structure factor by introducing an atomic mean square displacement tensor leads to the dynamic structure factors, which can be compared with the experimental ones (when properly corrected for secondary effects).

In spite of a non-negligible dispersion of experimental data, comparison of our theoretical results with experiment gives a satisfactory agreement. Moreover, the introduction of a very slight deviation of the F atom with respect to its ideal position still improves this agreement. However, the negative sign of the anharmonic vibration parameter β_F is not confirmed by the present results.

The precision of our calculations shows that the structure factors associated with the 200 and 222 reflections indicate a small but significant contraction

of the electron clouds, particularly for F^- , with respect to isolated ions.

Financial support from the Human Capital and Mobility Programme of the European Union under contract CHRX-CT93-0155 is gratefully acknowledged.

References

- ABRAHAMS, S. C., ALEXANDER, L. E., FURNAS, T. C., HAMILTON, W. C., LADELL, J., OKAYA, Y., YOUNG, R. A. & ZALKIN, A. (1967). *Acta Cryst.* **22**, 1-6.
- AZAVANT, P., LICHANOT, A., RÉRAT, M. & CHAILLET, M. (1994). *Theor. Chim. Acta*, **89**, 213-226.
- BACHMANN, R., KOHLER, H., SCHULZ, H. & WEBER, H. P. (1985). *Acta Cryst.* **A41**, 35-40.
- CATTI, M., DOVESI, R., PAVESE, A. & SAUNDERS, V. R. (1991). *J. Phys. Condens. Matter*, **3**, 4151-4164.
- CHEETHAM, A. K. & WILKINSON, A. P. (1992). *Angew. Chem.* **31**, 1557-1570.
- COOPER, M. J. (1970). *Acta Cryst.* **A26**, 208-214.
- DOVESI, R., PISANI, C., ROETTI, C., CAUSÀ, M. & SAUNDERS, V. R. (1989). *CRYSTAL88*, QCPE Program No. 577. Quantum Chemistry Program Exchange, Indiana Univ. Bloomington, IN, USA.
- DOVESI, R., ROETTI, C. & SAUNDERS, V. R. (1992). *CRYSTAL92*, User Documentation. Univ. of Torino, Italy, and SERC Daresbury Laboratory, England.
- ELCOMBE, M. M. & PRYOR, A. W. (1970). *J. Phys. C*, **3**, 492-499.
- FERRERO, E. (1981). Thesis, Univ. of Torino, Italy.
- International Tables for Crystallography* (1992). Vol. C, p. 206. Dordrecht: Kluwer Academic Publishers.
- MACKENZIE, J. K. & MASLEN, V. W. (1968). *Acta Cryst.* **A24**, 628-638.
- PISANI, C., DOVESI, R. & ORLANDO, R. (1992). *Int. J. Quantum Chem.* **42**, 5-33.
- PISANI, C., DOVESI, R. & ROETTI, C. (1988). *Hartree-Fock Ab Initio Treatment of Crystalline Solids. Lecture Notes in Chemistry Series*, No. 48. Berlin: Springer.
- RIECK, W., EULER, H. & SCHULZ, H. (1988). *Acta Cryst.* **A44**, 1099-1101.
- RIECK, W., SCHULZ, H. & SIEDEL, M. (1991). *J. Phys. Chem. Solids*, **52**, 1289-1291.
- SCHUMANN, B. & NEUMANN, H. (1984). *Cryst. Res. Technol.* **19**, K₁₃-K₁₄.
- SPACKMAN, M. A. (1986). *Acta Cryst.* **A42**, 271-281.
- TOGAWA, S. (1964). *J. Phys. Soc. Jpn.* **19**, 1696-1703.
- WHITE, G. K. (1980). *J. Phys. C*, **13**, 4905-4913.
- WILLIS, B. T. M. (1965). *Acta Cryst.* **18**, 75-76.
- WILLIS, B. T. M. & PRYOR, A. W. (1975). *Thermal Vibrations in Crystallography*, p. 96. Cambridge Univ. Press.
- WYCKOFF, W. G. (1968). *Crystal Structures*, Vol. 1, 2nd ed. New York: Wiley.
- ZACHARIASEN, W. H. (1968). *Acta Cryst.* **A24**, 425-427.